#### Interatomic potentials

What they can and cannot do

#### **Functional Forms**

- Must be such as to allow million atom MD
- Short-ranged (order-N calculation)

- Should describe electronic structure
- Motivated by DFT (a *sufficient* theory)
- Fitted to relevant properties

# History lesson

- 1930s Pair potentials  $\sum_{ii} V(r_{ii})$
- 1980s Many body potentials  $\sum_{ij} V(r_{ij}) + F_i[\sum_j \phi(r_{ij})]$
- 1990s Angle-dependence

$$\sum_{ij} V(r_{ij}) + F[\sum_{ij} \phi(r_{ij})] + G[\sum_{ijk} \Theta(r_{ij}, r_{jk}.\theta_{ijk})]$$

• 2000s Onsite dependence:

$$\sum_{i} Min_{\mu_i} [F(\mu_i, \sum_{j} \phi(r_{ij})) + \sum_{j} V_i(r_{ij}, \mu_i)]$$

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# What can empirical/MD do?

- Reproduce reliable energies YES
- Predict reliable energies NO
- Reproduce mechanisms and correlations YES
- Predict unexpected mechanisms YES

#### Potential developments



 $\alpha_i - \alpha_i / 2 = \alpha_i = \alpha_i + \alpha_i / 2$ 

### Two Band Model

- Allows localised orbitals on atom to change state – e.g. s-d or ↓ ↑
- Demonstrated for s-d transfer gives a discontinuous isostructural transition.
- Work in progress for iron
- Minimisation done analytically => still EAM-type speed not Car-Parrinello.

Cohesive term looks like Finnis-Sinclair with variable number of electrons in each band (second moment tight-binding)

$$U_{bond} = \sum_{i} \int_{-W_{i}/2}^{E_{f} = (\frac{n}{N} - \frac{1}{2})W_{i}} E \frac{N}{W_{i}} dE = -\sum_{i} \frac{W_{i}}{2N} n(N-n)$$

Plus extra energy associated with band centres being different

$$U_{bond} = \sum_{i} \frac{W_{i1}}{2N_1} n_{i1} (n_{i1} - N_1) + \frac{W_{i2}}{2N_2} n_{i2} (n_{i2} - N_2) + E_{prom}$$

Plus a pairwise potential (also dependent on occupation)

Problem: minimising *total* energy with respect to electron transfer at *each* site  $\eta_1$  appears to be minimisation of N-variable function

BUT, by writing energy as sum of atomic energies, and splitting the pair potential part between atoms appropriately, determining  $\eta_t$  becomes *local* and *analytic* (i.e. as fast as pair potentials) in the absence of charge transfer

$$\eta_{i_0} = \frac{N_1 N_2}{W_{i_1} N_2 + W_{i_2} N_1} \left[ W_{i_1} - W_{i_2} - T\left(\frac{W_{i_1}}{N_1} - \frac{W_{i_2}}{N_2}\right) + 2E_0 - 2U_{i_1, pair} + 2U_{i_2, pair} \right]$$

W and U are sums of pair potentials  $E_0$ , N and T are constants (band centre offset, band capacities, total number of electrons/atom)

#### BETTER STILL...

Force is variational in  $\eta$ , so can be evaluated locally (quickly) for MD. (Hellman-Feynman theorem!)

Elasticity is non-local – great flexibility in fitting.

$$\mathbf{f}_{i} = -\frac{dU_{tot}}{d\mathbf{r}_{i}}$$

$$= -\frac{\partial U_{tot}}{\partial \mathbf{r}_{i}}\Big|_{\eta} - \frac{\partial U_{tot}}{\partial \eta}\frac{\partial \eta}{d\mathbf{r}_{i}}$$

$$= -\frac{\partial U_{tot}}{\partial \mathbf{r}_{i}}\Big|_{\eta} \qquad (15)$$

$$\mathbf{f_i} = \sum_{j} \left[ (\tilde{W_{i1}} + \tilde{W_{j1}})\phi_1'(r_{ij}) + (\tilde{W_{i2}} + \tilde{W_{j2}})\phi_2'(r_{ij}) - (n_{i1} + n_{j1})\mathbf{V'}_1(r_{ij}) - (n_{i2} + n_{j2})\mathbf{V'}_2(r_{ij}) \right] \hat{\mathbf{r}}_{ij}$$

### Two band model – caesium

- s-d transfer
- Isostructural phase elastic constants (GPa) transition
- Elastic anomalies
- d-like atom is smaller – favoured under pressure
- Correlation effect beyond DFT-GGA!



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#### Electron transfer

• Optimise Ns-Nd

• Fit phase transition pressure, energy, volumes



#### Two-band model - transferability

 Parameters fitted to Cs describe all 6s5d metals!

• Suggests the physics is right



# Composition dependent potentials

- For alloys, properties depend on Fermi energy.
- Potentials capture only local effects (e.g. pinning), not global effects (phase stability)
- Two-band model shows that the total number of electrons can be included in fit *without* extra computational cost
- For MD, composition-dependent potentials can be generated at start of each run with no computing cost.

# Links Between the Different Project Tasks and Deliverables



# Role of MD in multiscale

- Identifying important processes
- Correlations
- Transformation mechanisms
- Trial defect geometries
- Key configurations for fitting to potential
- Not accurate energies from extrapolation these come from ab initio

### Crucial R&D needs

• Better understanding by modellers of the key materials problems.

• Commitment by experimentalists to think about their problems at the atomic level.